# Magnetic properties and magnetic entropy change in spinels $(Cd, M)Cr_2S_4$ with M=Cu or Fe

Jun Shen<sup>a)</sup>

State Key Laboratory of Magnetism, Beijing National Laboratory for Condensed Matter Physics and Institute of Physics, Chinese Academy of Sciences, Beijing 100080, People's Republic of China and School of Material Science and Engineering, Hebei University of Technology, Tianjin 300130, People's Republic of China

Li-Qin Yan, Jian Zhang, Fang-Wei Wang, Ji-Rong Sun, Feng-Xia Hu, and Chuan-Bing Rong

State Key Laboratory of Magnetism, Beijing National Laboratory for Condensed Matter Physics and Institute of Physics, Chinese Academy of Sciences, Beijing 100080, People's Republic of China

#### Yang-Xian Li

School of Material Science and Engineering, Hebei University of Technology, Tianjin 300130, People's Republic of China

(Presented on 6 November 2007; received 10 September 2007; accepted 17 October 2007; published online 29 January 2008)

Magnetic properties and magnetic entropy changes in spinels  $(Cd, M)Cr_2S_4$  with M=Cu or Fe have been studied. The saturation moments per formula unit are about  $4.77\mu_B$  for  $Cd_{0.8}Cu_{0.2}Cr_2S_4$  and  $4.41\mu_B$  for  $Cd_{0.7}Fe_{0.3}Cr_2S_4$ . Magnetic transition in Cu- and Fe-doped samples are of second order in nature, and their Curie temperatures are  $T_C=86$  and 119 K, respectively. The maximum values of magnetic entropy change  $\Delta S$  are found to be 5.1 and 5.4 J/kg K for  $Cd_{0.8}Cu_{0.2}Cr_2S_4$  and  $Cd_{0.7}Fe_{0.3}Cr_2S_4$  for a field change from 0 to 5 T, respectively. The significant  $\Delta S$  suggests a potential of the sulfospinel as a magnetic refrigerant in the corresponding temperature range. @ 2008 American Institute of Physics. [DOI: 10.1063/1.2830973]

#### **I. INTRODUCTION**

Room-temperature magnetic refrigeration based on the magnetocaloric effect (MCE) has been demonstrated to be a very promising alternative to conventional vapor-cycle refrigeration due to its potential impact on energy savings and environmental concerns.<sup>1–3</sup> Therefore, it has aroused the worldwide interest in the development of new magnetic refrigerants with a large MCE. A variety of prototype materials involving second-order and first-order magnetic transitions had been investigated theoretically and experimentally in an attempt to achieve a large MCE.<sup>4-10</sup> A first-order transition tends to concentrate magnetic entropy change  $\Delta S$  to a narrow temperature range around the transition temperature compared with second-order one, which implies that a large  $\Delta S$  is much easier to obtain in the vicinity of transition temperature. Although  $\Delta S$  of materials with second-order magnetic transition are generally lower than those with first-order one, they exhibit a good reversible behavior in the magnetization as functions of temperature and magnetic field, which is very useful for magnetic refrigeration applications.

Recently, the  $AB_2X_4$ -type sulfospinel has attracted much attention because of its colossal magnetocapacitive effects<sup>11</sup> (CMC) and large magnetoresistance effect<sup>12</sup> observed in CdCr<sub>2</sub>S<sub>4</sub>. Most of them are possessed of ferromagnetic spin ordering and large spontaneous magnetization. CdCr<sub>2</sub>S<sub>4</sub> exhibits a second-order magnetic transition at  $T_C$ =87 K with ferromagnetically coupled Cr<sup>3+</sup> spins (*S*=3/2).<sup>11</sup> Very recently, our studies revealed that CdCr<sub>2</sub>S<sub>4</sub> could also be a promising candidate for magnetic refrigerant because of its large MCE.<sup>13</sup> A maximum value of  $\Delta S$  is found to be 7 J/kg K for a field change of 0–5 T around its Curie temperature. In the present paper, we report on the effects of substitution of Cu and Fe for Cd in CdCr<sub>2</sub>S<sub>4</sub> on magnetic properties and magnetic entropy changes.

### **II. EXPERIMENTAL DETAILS**

Polycrystalline samples  $Cd_{0.8}Cu_{0.2}Cr_2S_4$  and  $Cd_{0.7}Fe_{0.3}Cr_2S_4$  were prepared by using the solid-state reaction method. A stoichiometric powder mixture of CdS(99.9), CuS(99.9), FeS(99.9), and  $Cr_2S_3(99\%)$  available from the Alfa Aesar Company was well ground, then pressed into pellets and sealed in evacuated quartz tubes. Afterward, these tubes were heated up to 780 °C and kept at this temperature for 4 days to obtain a single phase. X-ray diffraction (XRD) measurement was performed by using Cu  $K\alpha$  radiation to identify the phase and the crystal structure. Magnetic measurement was carried out by using a superconducting quantum interference device magnetometer.

## **III. RESULTS AND DISCUSSION**

Figure 1 shows the room-temperature powder XRD patterns for  $Cd_{0.8}Cu_{0.2}Cr_2S_4$  and  $Cd_{0.7}Fe_{0.3}Cr_2S_4$ . It can be found that  $Cd_{0.7}Fe_{0.3}Cr_2S_4$  crystallized in a very clean single phase of a normal spinel structure with the space group of Fd3m. For  $Cd_{0.8}Cu_{0.2}Cr_2S_4$ , all the diffraction peaks can be indexed by using the JADE 5.0 program<sup>14</sup> to a face-centercubic cell, except some smaller peaks (centered at about 33.6°, 36.3°, and 41.5°), which indicates the existence of a

0021-8979/2008/103(7)/07B315/3/\$23.00

103, 07B315-1

<sup>&</sup>lt;sup>a)</sup>Electronic mail: sj@g203.iphy.ac.cn



FIG. 1. Room-temperature powder XRD patterns for  $Cd_{0.8}Cu_{0.2}Cr_2S_4$  and  $Cd_{0.7}Fe_{0.3}Cr_2S_4$ . Bragg reflections (small vertical lines) are also shown.

minor  $Cr_2O_3$  as an impurity phase. The lattice parameters are found to be 10.246 Å for  $Cd_{0.8}Cu_{0.2}Cr_2S_4$  and 10.174 Å for  $Cd_{0.7}Fe_{0.3}Cr_2S_4$ .

Figure 2 shows the thermomagnetic M-T curves for Cd<sub>0.8</sub>Cu<sub>0.2</sub>Cr<sub>2</sub>S<sub>4</sub> and Cd<sub>0.7</sub>Fe<sub>0.3</sub>Cr<sub>2</sub>S<sub>4</sub> in a magnetic field of 0.01 T on heating and cooling. The Curie temperature  $T_C$  is defined as the temperature with a maximum slope in the M-Tcurves. The value of  $T_C$  for CdCr<sub>2</sub>S<sub>4</sub> is 86 K.<sup>13</sup> The partial replacement of Cd by Cu can exert a little influence on the magnetic coupling, and only a small shift of  $T_C$  from 86 to 88 K is observed. In contrast, a significant increase of  $T_C$  from 86 to 119 K is observed, which stems from the substitution of Fe for Cd. As shown in Fig. 2, the magnetic transition is completely reversible for the temperature increase-decrease cycling, which is indicative of the secondorder character of the magnetic transition. However, the magnetizations of Cd<sub>0.8</sub>Cu<sub>0.2</sub>Cr<sub>2</sub>S<sub>4</sub> and Cd<sub>0.7</sub>Fe<sub>0.3</sub>Cr<sub>2</sub>S<sub>4</sub> are strongly dependent on their thermal history at low temperatures. The heating and cooling magnetization curves each



FIG. 2. Temperature dependence of magnetization for  $Cd_{0.8}Cu_{0.2}Cr_2S_4$  and  $Cd_{0.7}Fe_{0.3}Cr_2S_4$  measured on heating and cooling in a magnetic field of 0.01 T. The insets show the magnetic field dependences of magnetization for  $Cd_{0.8}Cu_{0.2}Cr_2S_4$  and  $Cd_{0.7}Fe_{0.3}Cr_2S_4$  at 5 K, respectively.



FIG. 3. Isothermal magnetization curves for  $Cd_{0.8}Cu_{0.2}Cr_2S_4$  (a) and  $Cd_{0.7}Fe_{0.3}Cr_2S_4$  (b) around the Curie temperature in magnetic fields up to 5.0 T.

manifest a strong irreversibility below  $T_C$ . The magnetization curves at 5 K of  $Cd_{0.8}Cu_{0.2}Cr_2S_4$  and  $Cd_{0.7}Fe_{0.3}Cr_2S_4$  are shown in the insets of Fig. 2, respectively. The magnetization curves show that a full saturation state is achieved under a field of ~0.8 T. Saturation moments per formula unit are about 4.77 and  $4.41\mu_B$  for  $Cd_{0.8}Cu_{0.2}Cr_2S_4$  and  $Cd_{0.7}Fe_{0.3}Cr_2S_4$ , respectively, indicating the formation of a ferromagnetic spin configuration due to the Cr–S–Cr superexchange interaction.<sup>15</sup> The evaluated values of saturated moments for  $Cd_{0.8}Cu_{0.2}Cr_2S_4$  and  $Cd_{0.7}Fe_{0.3}Cr_2S_4$  are 5.8 and  $4.8\mu_B/f.u.$  based on the ideal  $Cu^{1+}$ ,  $Cr^{3+}$ ,  $Cr^{4+}$ , and  $Fe^{2+}$ ions.<sup>16</sup> The measured smaller values must arise from the formation of low spin state  $Cr^{2+}$  due to the sulfur loss in the sintering process.<sup>17</sup>

Figure 3 shows the isothermal magnetization curves for  $Cd_{0.8}Cu_{0.2}Cr_2S_4$  and  $Cd_{0.7}Fe_{0.3}Cr_2S_4$  in a wide temperature range around the Curie temperature with different temperature steps in magnetic fields up to 5.0 T. The temperature steps are chosen to be 2 and 5 K in the vicinity of  $T_C$  and in the regions far away from  $T_C$ , respectively. The sweep rate of the field is quite slow to ensure that the *M*-*H* curves are recorded in an isothermal process. Each isotherm shows a reversible behavior for the field ascending and descending and there is no inflection or negative slope that occurs in the Arrott plots as a feature of second-order phase transition.

The magnetic entropy change  $\Delta S$  is calculated from the Maxwell relation

$$\Delta S(T,H) = S(T,H) - S(T,0) = \int_0^H \left(\frac{\partial M}{\partial T}\right)_H dH$$

by using the isothermal magnetization data and can be evaluated from the expression

Downloaded 13 Sep 2008 to 159.226.36.179. Redistribution subject to AIP license or copyright; see http://jap.aip.org/jap/copyright.jsp



FIG. 4. Temperature dependence of the magnetic entropy change for  $Cd_{0.8}Cu_{0.2}Cr_2S_4$  (a) and  $Cd_{0.7}Fe_{0.3}Cr_2S_4$  (b) for magnetic field change from 0 to 2 T and from 0 to 5 T.

$$|\Delta S| = \sum_{i} \frac{M_i - M_{i+1}}{T_{i+1} - T_i} \Delta H_i.$$

Figure 4 shows the values of  $\Delta S$  for Cd<sub>0.8</sub>Cu<sub>0.2</sub>Cr<sub>2</sub>S<sub>4</sub> and  $Cd_{0.7}Fe_{0.3}Cr_2S_4$  as functions of temperature for a field change from 0 to 2 T and from 0 to 5 T, respectively. Around Curie temperatures, a negative peak can be observed from the curves and the maximum values of  $\Delta S$  for a magnetic field change from 0 to 5 T are found to be 5.1 and 5.4 J/Kg K, respectively. The values of  $\Delta S$  in the present samples exhibit a small decrease compared with those of  $CdCr_2S_4$  (Ref. 13) due to the decrease of magnetization caused by the addition of Cu and Fe. Although the maximum  $\Delta S$  in CdCr<sub>2</sub>S<sub>4</sub> based spinels is smaller than that of the most conspicuous magnetocaloric material,  $^{6-10}$  the  $\Delta S$  distribution is much more uniform than that of typical first-order phase transition, which is desirable for an Ericsson-cycle magnetic refrigerator.<sup>18</sup> For  $Cd_{0.8}Cu_{0.2}Cr_2S_4$  and  $Cd_{0.7}Fe_{0.3}Cr_2S_4$ , the widths of the  $\Delta S$ peak,  $\delta T_{\rm FWHM}$ , defined as the temperature interval corresponding to the half maximum of  $\Delta S$ , approach to ~55 and  $\sim$ 46 K, respectively, in the magnetic field change from 0 to 5 T. According to the mean field theory, the relation between magnetic entropy change and the magnetic field near the Curie temperature is described as<sup>19</sup>

$$\Delta S \approx -1.07 q R \left(\frac{g \mu_B J H}{k T_c}\right)^{2/3},$$

where *q* is the number of magnetic ions, *R* is the gas constant, and *g* is the Laude factor. Figure 5 shows the  $H^{2/3}$  dependence of the  $\Delta S$  for  $Cd_{0.8}Cu_{0.2}Cr_2S_4$  and  $Cd_{0.7}Fe_{0.3}Cr_2S_4$ .

It is found that the  $\Delta S$  well linearly depends on the  $H^{2/3}$  near Curie temperature, implying the strong localization of 3d electrons and the second-order character of magnetic transition.<sup>20</sup> The value of isothermal magnetic entropy change will rise with the increase of magnetic field. A large  $\Delta S$  for the present samples results from the sharp drop in magnetization as the temperature passes through  $T_c$ .

In summary, we have investigated the effects of the substitution of Cu and Fe for Cd on magnetic properties and



FIG. 5. (Color online) Relation between maximum  $\Delta S$  and  $H^{2/3}$  near Curie temperature for Cd<sub>0.8</sub>Cu<sub>0.2</sub>Cr<sub>2</sub>S<sub>4</sub> and Cd<sub>0.7</sub>Fe<sub>0.3</sub>Cr<sub>2</sub>S<sub>4</sub>. The straight line gives the linear fitted to data.

magnetic entropy changes, based on the discovery of large magnetocaloric effect in CdCr<sub>2</sub>S<sub>4</sub>.<sup>13</sup> It is found that the substitution of Cu or Fe for Cd in CdCr<sub>2</sub>S<sub>4</sub> may lead to a decrease in the spontaneous magnetization and an increase in the Curie temperature. The large values of  $\Delta S$  for Cd<sub>0.8</sub>Cu<sub>0.2</sub>Cr<sub>2</sub>S<sub>4</sub> and Cd<sub>0.7</sub>Fe<sub>0.3</sub>Cr<sub>2</sub>S<sub>4</sub> compounds with a second-order magnetic field change of 0–5 T. A reversible behavior of the magnetization with temperature and magnetic field and a broad distribution of  $\Delta S$  peak have been achieved.

The present work has been supported by the National Basic Research Program of China, the National Natural Science Foundation of China, and the Basic Research Program of Chinese Academy of Sciences.

- <sup>1</sup>C. B. Zimm, A. Jastrab, A. Sternberg, V. K. Pecharsky, K. A. Gschneidner, Jr., M. Osborne, and I. Anderson, Adv. Cryog. Eng. **43**, 1759 (1998).
- <sup>2</sup>A. M. Tishin and Y. I. Spichkin, in *The Magnetocaloric Effect and its Applications*, edited by J. M. D. Coey *et al.* (OP, Bristol, 2003).
- <sup>3</sup>K. A. Gschneidner, Jr., V. K. Pecharsky, and A. O. Tsokol, Rep. Prog. Phys. **68**, 1479 (2005).
- <sup>4</sup>A. M. Tishin, *Handbook of Magnetic Materials*, edited by K. H. J. Buschow (North-Holland, Amsterdam, 1999), Vol. 12, p. 395.
- <sup>5</sup>V. K. Pecharsky and K. A. Gschneidner, Jr., J. Magn. Magn. Mater. **200**, 44 (1999).
- <sup>6</sup>V. K. Pecharsky and K. A. Gschneidner, Jr., Phys. Rev. Lett. **78**, 4494 (1997).
- <sup>7</sup>F. X. Hu, B. G. Shen, J. R. Sun, and X. X. Zhang, Chin. Phys. **9**, 550 (2000).
- <sup>8</sup>F. X. Hu, B. G. Shen, J. R. Sun, Z. H. Chen, G. H. Rao, and X. X. Zhang, Appl. Phys. Lett. **78**, 3675 (2001).
- <sup>9</sup>H. Wada and Y. Tanabe, Appl. Phys. Lett. **79**, 3302 (2001).
- <sup>10</sup>O. Tegus, E. Brück, K. H. J. Buschow, and F. R. de Boer, Nature (London) **415**, 150 (2002).
- <sup>11</sup>J. Hemberger, P. Lunkenheimer, R. Fichtl, H.-A. Krug von Nidda, V. Tsurkan, and A. Loidl, Nature (London) **434**, 364 (2005).
- <sup>12</sup>H. W. Lehmann and M. Robbins, J. Appl. Phys. **37**, 1389 (1966).
- <sup>13</sup>L. Q. Yan, J. Shen, Y. X. Li, F. W. Wang, Z. W. Jiang, F. X. Hu, J. R. Sun, and B. G. Shen, Appl. Phys. Lett. **90**, 262502 (2007).
- <sup>14</sup>Materials Data Inc. 1999 JADE 5.0 XRD Pattern Processing.
- <sup>15</sup>P. K. Baltzer, P. J. Wojtowicz, M. Robbins, and E. Lopatin, Phys. Rev. 151, 367 (1966).
- <sup>16</sup>J. Krok-Kowalski, T. Grofi, J. Warczewski, T. Mydlarz, and I. Okofiska-Koztowska, J. Magn. Magn. Mater. **168**, 129 (1997).
- <sup>17</sup>G. Haack and L. C. Beegle, J. Appl. Phys. **39**, 656 (1968).
- <sup>18</sup>G. V. Brown, J. Appl. Phys. 47, 3673 (1976).
- <sup>19</sup>H. Oesterriecher and F. T. Parker, J. Appl. Phys. 55, 4334 (1984).
- <sup>20</sup>Q. Y. Dong, H. W. Zhang, J. L. Shen, J. R. Sun, and B. G. Shen, J. Magn. Magn. Mater. **319**, 56 (2007).